

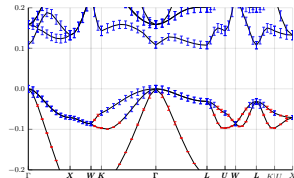
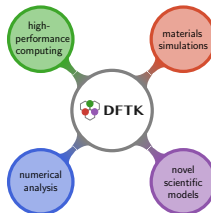
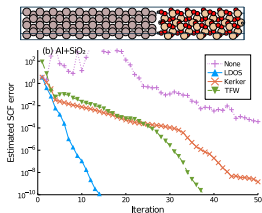
# Towards error-controlled, black-box density-functional theory methods

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Slides: <https://michael-herbst.com/slides/vmd2021>



# Motivation: Computational challenges

- Virtual materials design  $\Rightarrow$  millions of calculations:
  - design space search, data generation for surrogates, ...
- Key requirements:
  - Automation (saves human time)
  - Efficiency (saves computer time)
  - **Reliability** (saves computer time & human time)
- State of the art:
  - Many parameters to choose (algorithms, tolerances, models)
  - Choice **by experience**
  - Workflow **success rate**:  $\simeq 50\%$ <sup>1</sup>
  - Little rigorous error control (basically trial and error)

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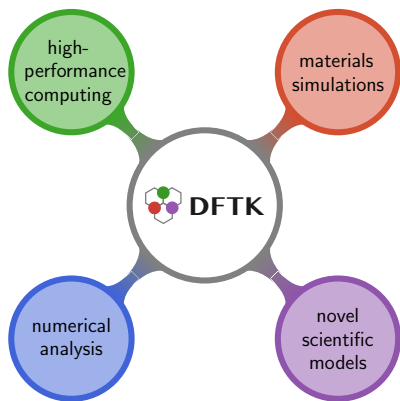
<sup>1</sup>Z. Ulissi, private communication in ARPAE differentiate group seminar, Dec 2020.

# Some questions and hardly explored techniques

- Leading questions:
  - Do we need the **same accuracy everywhere?**
  - Where can mathematical insight **improve reliability?**
  - Can we tune between accuracy and runtime using only a **single parameter?**
- Error estimation (*a posteriori*, UQ, sensitivity analysis)
  - ⇒ Error-guided automatic selection of parameters
- Numerical analysis of simulation algorithms
  - ⇒ Towards black-box algorithms
- Multi-fidelity methods
  - ⇒ Combination of results of different quality (functionals, numerics, corrections, ...)

⇒ **Multidisciplinary research setting**

# Density-functional toolkit (DFTK)<sup>1</sup>



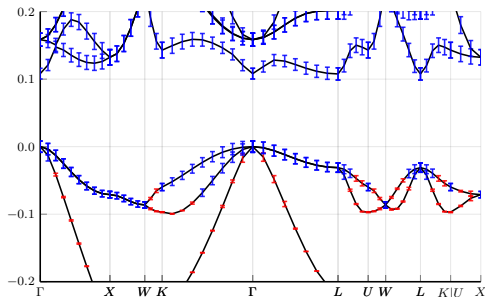
- <https://dftk.org>
- 2 years of development
- Open-source **julia** code
- Building on **julia** ecosystem
- Supports **mathematical developments** *and* scale-up to regime relevant to **applications**
- Low entrance barrier: Only 6k lines of code!

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<sup>1</sup>M. F. Herbst, A. Levitt and E. Cancès. JuliaCon Proc., 3, 69 (2021).



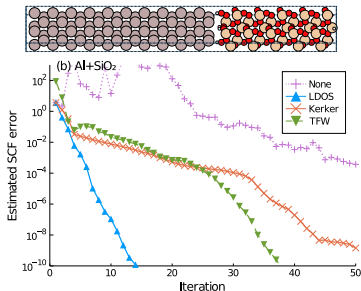
- Documentation and examples: <https://docs.dftk.org>
- Ground state (LDA, GGA) and a bit of response theory
- Compose your model (e.g. analytic potentials, 1D / 2D, ...)
- Arbitrary floating point type / reduced precision
- Automatic differentiation (e.g. stresses, sensitivities)
- Mixed MPI-Thread-based parallelism
- > 800 electrons possible
- **Performance:** Within factor 2–4 of established codes
- Involved in multiple **multidisciplinary collaborations:**
  - ARPA-E's ACED-differentiate, ERC's EMC2 synergy,  
MIT's Center for the Exascale Simulation of Material Interfaces (CESMIX)



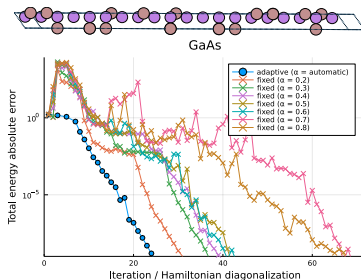
- Reduced model *without SCF*, otherwise representative setting
- **Error bars guaranteed**: Difference to analytical solution
- Hint what to improve: Tolerance, basis, floating-point type
- Just a starting point . . .
- Model error: Combination of analytical and statistical approaches (BEEF)

<sup>1</sup>M. F. Herbst, A. Levitt and E. Cancès. Faraday Discuss. **224**, 227 (2020).

# Black-box SCF methods: Our recent ideas



- Parameter-free mixing for inhomogeneous systems<sup>1</sup>



- Adaptive and automatic selection of the damping parameter

<sup>1</sup>M. F. Herbst, A. Levitt. J. Phys. Condens. Matter **33**, 085503 (2021).

# Self-consistent field (SCF) as a fixed-point problem

- Self-consistent field equations:

$$\begin{cases} \hat{\mathcal{F}}_{\rho} = -\frac{1}{2}\Delta + \mathcal{V}(\rho) \\ \rho(\underline{\mathbf{r}}) = \sum_i f\left(\frac{\varepsilon_i - \varepsilon_F}{T}\right) |\psi_i(\underline{\mathbf{r}})|^2 \quad \text{with } \hat{\mathcal{F}}_{\rho}\psi_i = \varepsilon_i\psi_i, \end{cases}$$

⇒ Fixed-point problem  $\rho = F(\rho)$

- Use damped update with **mixing** (preconditioner)  $P$ :

$$\rho_{n+1} = \rho_n + \alpha P^{-1} [F(\rho_n) - \rho_n]$$

- Near a fixed-point the error goes as

$$e_{n+1} \simeq [1 - \alpha P^{-1} \epsilon^{\dagger}] e_n$$

with  $\epsilon^{\dagger}$  dielectric matrix

- Convergence iff  $-1 < [1 - \alpha P^{-1} \epsilon^{\dagger}] < 1$

⇒ Need  $P^{-1} \simeq (\epsilon^{\dagger})^{-1}$  (**matching preconditioner**) or **small**  $\alpha$

⇒  $\kappa(P^{-1} \epsilon^{\dagger})$  determines convergence rate

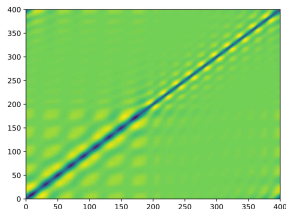


# Drawback of established approaches

1. Ideal mixing  $P$  is system-dependent, but manually chosen
    - Rough idea of dielectric properties needed *a priori*
    - Good preconditioners only known for bulk materials
    - Misses important applications (e.g. inhomogeneous systems)
    - Examples: Metal clusters, passivated surfaces, heterogeneous catalysis, ...
  2. No good mixing  $P$  known
    - Damping  $\alpha$  found by trial and error
- Our motivation: Making these cases more black-box

# Solving 1: LDOS mixing<sup>1</sup>

- Bulk preconditioning models tackle directly  $P^{-1} \approx (\epsilon^\dagger)^{-1}$
- But we have  $\epsilon^\dagger = (1 - \chi_0 K_{\text{Hxc}})$
- Plot of  $\chi_0$  (Chain of 10 Sodium atoms and 10 helium atoms):



⇒ Diagonal-dominant, try to approx.  $\chi_0(\underline{r}, \underline{r}') \simeq \widetilde{\chi}_0(\underline{r})\delta(\underline{r}, \underline{r}')$ :

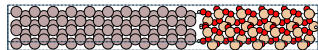
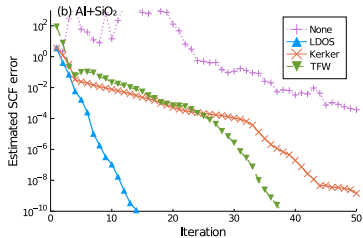
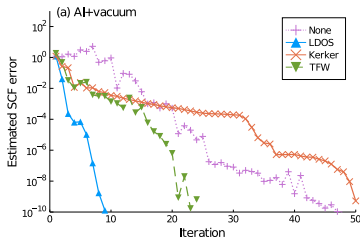
$$P^{-1}\rho_n = (1 - \widetilde{\chi}_0 K_{\text{Hxc}})^{-1} \rho_n \quad (\text{iteratively})$$

- In the case of **LDOS mixing**:

$$\chi_0(\underline{r}, \underline{r}') \simeq -\text{LDOS}(\underline{r})\delta(\underline{r}, \underline{r}')$$

<sup>1</sup>M. F. Herbst, A. Levitt. J. Phys. Condens. Matter **33**, 085503 (2021).

# LDOS preconditioning (examples)<sup>2</sup>



- 20 repeats of aluminium + 20 repeats vacuum / silica
- **TFW**: local Thomas-Fermi-von Weizsäcker mixing<sup>1</sup>
- **LDOS** automatically interpolates between Kerker mixing (in the metallic region) and no mixing (insulating region)

⇒ Parameter-free and black-box

<sup>1</sup>D. Raczowski, A. Canning, L. W. Wang, Phys. Rev. B. **64**, 121101 (2001).

<sup>2</sup>M. F. Herbst, A. Levitt. J. Phys. Condens. Matter **33**, 085503 (2021).

## Solving 2: Adaptive damping

- Potential mixing:

$$V_{n+1} = V_n + \alpha \delta V_n$$
$$\delta V_n = P^{-1} [\mathcal{V}(\mathcal{D}(V_n)) - V_n]$$

- Quadratic model for DFT energy:

$$E(V_n + \alpha \delta V_n) \simeq E(V_n) + \alpha \left\langle \nabla E|_{V=V_n} \middle| \delta V_n \right\rangle$$
$$+ \frac{\alpha^2}{2} \left\langle \delta V_n \middle| \nabla^2 E|_{V=V_n} \delta V_n \right\rangle$$

- After some algebra:

$$\nabla E|_{V=V_n} = -\chi_0 [\mathcal{V}(\mathcal{D}(V_n)) - V_n]$$
$$\nabla^2 E|_{V=V_n} \simeq -\chi_0 [1 - K_{\text{Hxc}} \chi_0]$$

⇒ Use model to find damping automatically!

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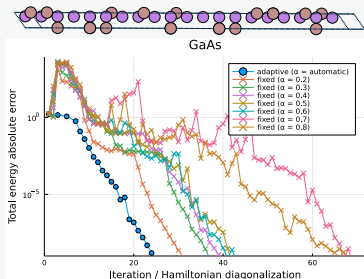
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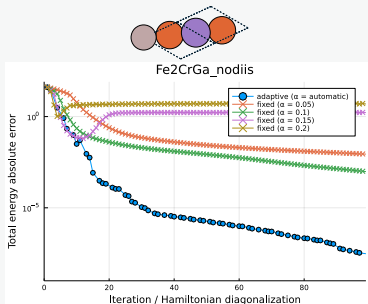
# Adaptive damping

## GaAs (with Anderson)



- No preconditioner ( $P = I$ )
- Non-linear SCF behaviour in initial steps

## Fe<sub>2</sub>CrGa Heusler (without Anderson)

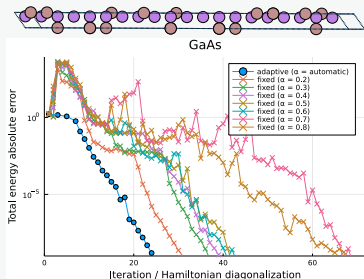


- Unsuitable Kerker preconditioner
- Localised states, spin

- Adaptive damping as black-box safeguard
- Ensures energy / residual decrease
- Interplay with Anderson tricky to interpret

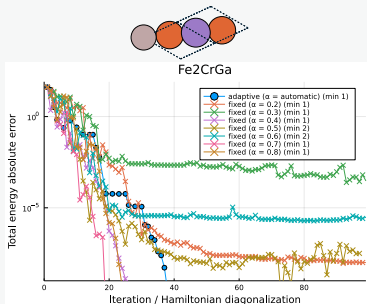
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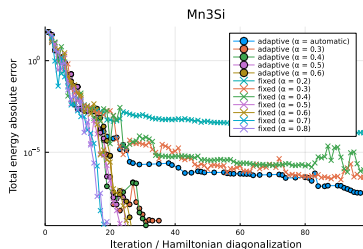
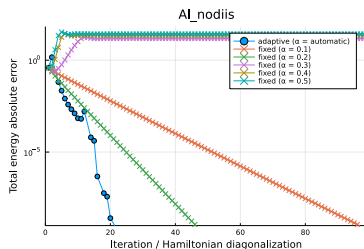
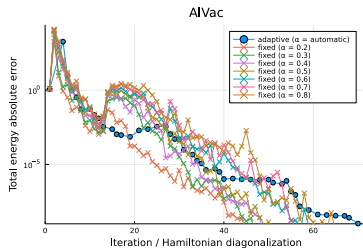
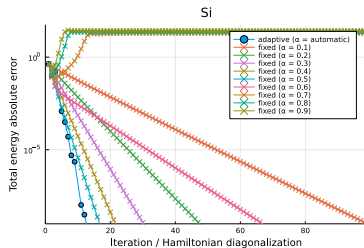
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- Unsuitable Kerker preconditioner
- Localised states, spin

- Adaptive damping as black-box safeguard
- Ensures energy / residual decrease
- Interplay with Anderson tricky to interpret

# Adaptive damping





-  **DFTK** usage:
  - First develop LDOS scheme on test systems (1D, toy problems)
  - Tests on  $> 800$  electrons (in the same code!)
  - Key quantities ( $\chi_0$ ,  $K_{xc}$ ) fully accessible
- Towards error-controlled methods:
  - Mathematically guided numerics matching the error of models
- LDOS preconditioner:
  - Parameter-free  $\Rightarrow$  Highly suitable for high-throughput
  - Adaptive preconditioning for inhomogeneous systems
- Adaptive damping scheme:
  - Safe guard if preconditioner not perfect / tricky system
  - Reduction of the human factor



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all DFTK contributors





**DFTK** <https://dftk.org>



**julia:** <https://michael-herbst.com/learn-julia>



**mfherbst**



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# Self-consistent field (SCF) as a fixed-point problem

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- Potential-to-density map  $\mathcal{D}$

$$\mathcal{D}(V) = \sum_{i=1}^{\infty} f\left(\frac{\varepsilon_i - \varepsilon_F}{T}\right) |\psi_i|^2$$

with  $(\varepsilon_i, \psi_i)$  eigenpairs of  $-\frac{1}{2}\Delta + V$ .

$\Rightarrow$  Fixed-point problem

$$\rho = \mathcal{D}(\mathcal{V}(\rho))$$

# Analysis of SCF convergence

- Fixed-point problem  $\rho = \mathcal{D}(\mathcal{V}(\rho)) \Rightarrow$  Use damped update

$$\rho_{n+1} = \rho_n + \alpha P^{-1} [\mathcal{D}(\mathcal{V}(\rho_n)) - \rho_n]$$

with **preconditioner** (“mixing”)  $P$

- Near a fixed-point the error goes as

$$e_{n+1} \simeq \left[ 1 - \alpha P^{-1} \epsilon^\dagger \right] e_n$$

where  $\epsilon^\dagger = 1 - \chi_0 K_{\text{Hxc}}$  (dielectric matrix)

- $\chi_0$ : Susceptibility (derivative of  $\mathcal{D}$ ),  $K_{\text{Hxc}}$ : Kernel (deriv. of  $\mathcal{V}$ )
- Convergence iff  $-1 < \left[ 1 - \alpha P^{-1} \epsilon^\dagger \right] < 1$ 
  - $\Rightarrow$  Need  $P^{-1} \simeq (\epsilon^\dagger)^{-1}$  (**matching preconditioner**) or **small**  $\alpha$
  - $\Rightarrow \kappa(P^{-1} \epsilon^\dagger)$  determines convergence rate

# SCF instabilities

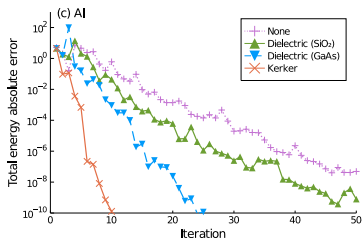
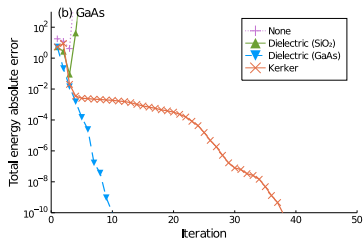
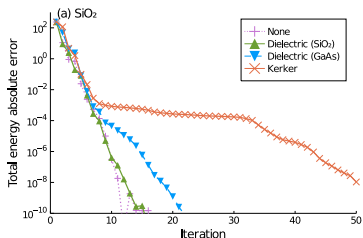
$$e_{n+1} \simeq [1 - \alpha P^{-1} \epsilon^\dagger] e_n, \quad \epsilon^\dagger = 1 - \chi_0(v_C + K_{xc})$$

- SCF instabilities increase condition number  $\kappa$ :
  - $\epsilon^\dagger$  has small eigenvalues (e.g. symmetry breaking)
  - $\chi_0$  has large eigenvalues (localised states)
  - Large charge-sloshing modes of  $v_C$  are uncompensated by  $\chi_0$  (metals)

⇒ Need infeasibly small  $\alpha$  or good  $P$

- Physics where a good mixing  $P$  is known:
  - *Bulk* insulators ( $P = I$ )
  - *Bulk* metals (Kerker mixing)
  - *Bulk* semiconductors (e.g. Resta's dielectric model)

# Convergence results for bulk materials<sup>1</sup>



- silica (SiO<sub>2</sub>) insulator
- gallium arsenide (GaAs) semiconductor
- aluminium (Al) metal

<sup>1</sup>M. F. Herbst, A. Levitt. J. Phys. Condens. Matter **33**, 085503 (2021).



# Local density of states (LDOS) approximation for $\chi_0^1$

- Main interest: Large-scale variations from  $\rho_n$  to  $\rho_{n+1}$
- $\Rightarrow$  Assume  $\underline{r} \mapsto \chi_0(\underline{r}, \underline{r}')$  more localised around  $\underline{r}'$  than  $V(\underline{r}')$ .
- “Row-sum mass lumping”:

$$\begin{aligned}\int \chi_0(\underline{r}, \underline{r}') V(\underline{r}') d\underline{r}' &\simeq V(\underline{r}) \int \chi_0(\underline{r}, \underline{r}') d\underline{r}' \\ &= -V(\underline{r}) D_{\text{loc}}(\underline{r})\end{aligned}$$

with local density of states

$$D_{\text{loc}}(\underline{r}) = \frac{1}{T} \sum_i f' \left( \frac{\varepsilon_i - \varepsilon_F}{T} \right) |\psi_i(\underline{r})|^2$$

using Adler-Wiser formula.

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# LDOS preconditioning results<sup>1</sup>

		None		Dielectric		Kerker		LDOS		LDOS+ Dielectric	
	$\mathcal{N}$	it	$\kappa$	it	$\kappa$	it	$\kappa$	it	$\kappa$	it	$\kappa$
SiO <sub>2</sub> +vacuum	10	11	3.3	26	19.7	50	95.7	11	3.3	26	19.7
	20	12	3.4	30	24.4	n.c.	351.5	12	3.4	30	21.7
GaAs+vacuum	10	17	13.4	18	6.2	23	67.0	17	12.4	18	10.4
	20	20	15.5	22	12.9	n.c.	312.2	20	15.5	22	12.9
Al+vacuum	10	19	51.5	24	44.3	22	64.4	9	3.7	16	10.3
	20	47	170.8	49	168.5	n.c.	323.9	9	3.5	20	10.5
GaAs+SiO <sub>2</sub> <sup>a</sup>	10	45	13.7	19	8.9	34	52.4	45	13.4	19	8.8
	20	n.c.	18.2	20	10.2	n.c.	170.1	n.c.	18.2	20	10.2
Al+SiO <sub>2</sub>	10	43	93.1	29	33.6	30	50.9	17	6.1	20	9.2
	20	n.c.	316.6	n.c.	118.4	n.c.	159.4	14	5.4	20	10.1
Al+GaAs	10	n.c.	144.0	24	22.4	16	9.0	15	7.2	11	3.5
	20	n.c.	485.0	40	59.0	26	28.8	26	21.4	13	5.0
Al+GaAs+SiO <sub>2</sub>	10	n.c.	149.5	34	50.4	36	62.9	26	21.5	19	9.0

- Coloured: Condition number  $\kappa$  less than doubled on doubling system size

<sup>1</sup>M. F. Herbst, A. Levitt. J. Phys. Condens. Matter **33**, 085503 (2021).